



The Improvement of Ion Plated Ag and Au Film Adherence to Si_3N_4 and SiC Surfaces for Increased Tribological Performance

Talivaldis Spalvins
Lewis Research Center, Cleveland, Ohio

National Aeronautics and
Space Administration

Lewis Research Center

Acknowledgments

The author would like to thank Dr. D. Wheeler for the XPS analysis.

Available from

NASA Center for Aerospace Information
7121 Standard Drive
Hanover, MD 21076
Price Code: A03

National Technical Information Service
5287 Port Royal Road
Springfield, VA 22100
Price Code: A03

The Improvement of Ion Plated Ag and Au Film Adherence to Si_3N_4 and SiC Surfaces for Increased Tribological Performance

T. Spalvins
National Aeronautics and Space Administration
Lewis Research Center
Cleveland, Ohio 44135

Abstract

A modified dc-diode plating system, utilizing a metallic screen cage as a cathode and referred as SCREEN CAGE ION PLATING (SCIP), is used to deposit Ag and Au lubricating films on Si_3N_4 and SiC surfaces. When deposition is performed in Ar or N_2 glow discharge, the surface displays poor adhesive strength (<5 MPa). A dramatic increase in adhesive strength (>80 MPa) is achieved when plating is performed in a reactive 50% O_2 + 50% Ar glow discharge. The excited/ionized oxygen species (O_2^+/O^+) in the glow discharge contribute to the oxidation of the Si_3N_4 or SiC surfaces as determined by X-ray Photoelectron Spectroscopy (XPS) depth profiling. The reactively sputter-oxidized Si_3N_4 or SiC surfaces and the activated-oxidized-metastable Ag or Au species formed in the plasma cooperatively contribute to the increased adherence. As a result, the linear thermal expansion coefficient mismatch at the interface is reduced. These lubricating Ag and Au films under sliding conditions reduce the friction coefficient by a factor of 2-1/2 to 4.

Introduction

Of all the various nonoxide ceramics used in engineering applications, Si_3N_4 and SiC are the prime candidate materials specified for the manufacture of mechanical components in advanced aircraft engines. Ceramic elements such as ball bearings, valve components, piston rings and cylinder liners are increasingly being specified for engines which have to withstand severe service operating conditions such as high temperatures, loads and speeds. Si_3N_4 has emerged as the best suited ceramic material for ball bearings, because of its high strength, hardness and relatively high toughness. But before widespread use of Si_3N_4 bearings can occur, the tribological performance must be significantly improved. Since ceramics are brittle materials, the presence of surface or near surface microflows is very detrimental to their performance. Ceramic surfaces which are in sliding, rolling, or oscillating motion generate tensile stresses, which are localized near the surface. These tensile stresses interact with the surface/subsurface flaws/defects and result in microcracking and chipping of the surface, which lead to severe wear.

It has been shown that adherent, soft Ag or Au films not only reduce the coefficient of friction during sliding contact but also reduce the surface tensile stresses that contribute to crack initiation and propagation and lead to microcracking and severe wear. In solid film lubrication the foremost requirement of the soft, metallic films is strong adherence, since the degree of adherence determines the durability of the film. Inherently, Ag and Au films display poor adherence to ceramic surfaces and cannot be directly deposited. To overcome this deficiency Ti or Cr interlayers have been used,¹⁻³ to achieve closer compatibility by decreasing the large difference in the Coefficient of Thermal Expansion (CTE) between the film and the substrate. The general rule is that the CTE of the coating should differ by less than 25 percent from that of the substrate.⁴ In the present study, the adhesion problem is solved by using the Oxygen assisted Screen Cage Ion

Plating (SCIP) process, which favorably affects the metal/ceramic interface and dramatically increases film adherence.

Concepts of Ceramic Lubrication

Since ceramics are processed from powders using sintering, hot-pressing or hot-isostatic pressing, they possess less than theoretical density. The resulting microstructures from these powder processes generally have some inherent nanoporosity. All these factors affect deformation/fracture behavior. As these ceramic surfaces come in sliding or rolling contact, high tensile stresses develop at asperities and interact with surface/subsurface flaws, thus initiating microcracks and fragmentation of the contact surface. Ceramics are weak in tension and have hardly any ductility. Thus even the slightest degree of porosity trapped within a grain is believed to be a source of failure.

Unlike metals, which yield locally when the elastic limit is exceeded, gross microfracture can occur, which is a typical wear mode for unlubricated ceramics in sliding.⁷ However, the load at which microfracture occurs can be increased by Ag or Au lubricating films.

According to the adhesion theory of friction,^{6,7} the frictional force, F , is proportional to the shear strength, s , and the real area of contact, A , ($F = As$) as shown in Fig. 1(a). For friction to be low, both A and s must be small. This means that the most suitable materials must have high hardness and low shear strength. However, this is not achievable with monolithic ceramics, therefore, an adherent, thin, ductile film should be interposed between the contacting surfaces as shown in Fig. 1(b).

The soft Ag and Au films shear easily and prevent opposing asperities from coming into frequent contact. This easy shear at contact interfaces produces fewer asperity/asperity interactions which result in lower friction and reduced wear. When friction is low, the magnitude of tensile stresses developing behind the moving asperities is also reduced. As a result the surficial tensile stresses that are responsible for crack initiation and propagation at the surface/subsurface are reduced. Lowering the friction coefficient significantly increases the critical load for the onset of crack initiation and formation.

Experimental Conditions

Specimens

In this investigation pure Si_3N_4 disks, 1.9 cm in diameter, and SiC square flats (1.2 cm x 1.2 cm) both with as surface finish of 0.1 μm rms, were plated with 0.2 to 0.4 μm thick Ag or Au films. Adhesion tests were performed on these coated specimens. Also, Si_3N_4 and SiC disks, 6.35 cm in diameter and surface finish of 0.2 μm rms, were coated with 1-3 μm thick Ag or Au films. The coated disks were tested under atmospheric conditions at 25 °C in a pin-on-disk tribometer.

Screen Cage Ion Plating (SCIP) of Ag and Au

The SCIP apparatus has been described in previous publications,^{8,9} its schematic is shown in Fig. 2. Briefly, it consists of a dc-diode configuration. The ceramic specimen is suspended in a screen cage (either silver or stainless steel plated with gold) that functions as both an electron grid and a cathode, while the resistance evaporative source acts as an anode. In the present study three

different gas compositions, pure Ar, pure N₂, and a mixture of 50% Ar + 50% O₂ are used. The total gas pressure for the generation of the plasma is set at a constant value of 20 mtorr. Before Ag or Au plating is performed at -4 kV, 80 mA, the specimen is soaked in the selected plasma at -2 kV, 20 mA for about 2 min. Because of the high throwing power of ion plating, all surfaces of the ceramic pieces are coated.

Adhesion Tests

A commercial pull-off adhesion tester (Sebastian IV) is used to determine the adhesive strength of the Ag and Au films to the Si₃N₄ and SiC substrates. In these tests, an epoxy coated Al pin is secured with a mounting spring clip to the film and transferred to an oven for curing at 150 °C for 1 hr. After curing, the test pin with the coated specimen is inserted into the instrument and the pin is pulled with an increasing force normal to the plane of the film until failure occurs. This test is limited by the nominal strength of the epoxy, which is about 80 MPa. When the adhesion of the film to the substrate is high and it is not possible to pull the film off, the epoxy breaks first.

Friction Testing

The Si₃N₄ and SiC disks, SCIP'd in (Ar+O₂) glow discharge, were friction tested with bare Si₃N₄ and SiC pins which were hemispherically tipped with a 2.54 cm radius of curvature. The pin-on-disk testing was performed in a high temperature tribometer.¹⁰ The tribotesting was performed under atmospheric conditions under 4.9N load at 1 m/sec (370 rpm) velocity for 30 min at 25 °C.

Results and Discussion

Adhesion and Coatings/Substrate Characterization

The adhesive pull-off tests were performed for the Ag and Au films plated in N₂, Ar, and 50% Ar + 50% O₂ plasmas on Si₃N₄ and SiC substrates at identical plating conditions. Usually, six pull-off tests were made on six separately plated specimens to determine the average adhesive strength as shown in Figs. 3(a) and (b). These results reveal that Ag and Au films plated in N₂ and Ar plasma on Si₃N₄ and SiC surface display poor adhesive strengths (<5 MPa). They can be detached by a scotch tape pull test. However, the adhesive strength is dramatically increased when the films are plated in a 50% Ar + 50% O₂ plasma. The oxygen-assisted deposition remarkably enhances the Ag and Au adhesive strength (>70 MPa) to the Si₃N₄ and SiC substrates. For the pull tests performed on specimens plated in Ar-O₂ plasma the failure occurred in the epoxy, indicating that the film adhesive strength is higher than the epoxy strength. The increased adhesive strength results clearly show that oxygen in the excited or ionized state (O₂⁺/O⁺) is instrumental in increasing the adhesive strength of the Ag and Au films to the Si₃N₄ and SiC substrates

Since the Si₃N₄ and SiC surfaces prior to Ag and Au deposition are exposed to the oxidizing (Ar + O₂) plasma for about 2 to 3 min, prior to plating these surfaces were analyzed by X-ray Photoelectron Spectroscopy (XPS). XPS depth profiles of SiC after Ar sputter etching in the non-oxidizing plasma and sputter etching in the (Ar + O₂) oxidizing plasma are shown in Figs. 4(a) and (b). XPS depth profiles of Si₃N₄ after Ar sputter etching and (Ar + O₂) sputter etching are shown in Figs. 5(a) and (b). These XPS depth profiles show the relative proportions of O, N, Si for Si₃N₄ and O, C, Si for SiC. From these composition profiles it is evident that the outermost Si₃N₄ and SiC layers are oxidized in the (Ar + O₂) plasma. The following surface oxidation reactions are proposed:^{11,12}

- (1) $\text{Si}_3\text{N}_4 \rightarrow \text{SiO}_x\text{N}_y \rightarrow \text{SiO}_2$
- (2) $\text{SiC} \rightarrow \text{SiC:O}_x$

Further, when these oxidized Si_3N_4 and SiC surfaces are then Ag and Au plated in an Ar or N_2 plasma, the adherence is poor, but when plated in $(\text{Ar} + \text{O}_2)$ plasma, the adherence is >70 MPa as shown in Figs. 3(a) and (b). The poor adherence of Ag and Au plated in Ar or N_2 plasma on the oxidized Si_3N_4 and SiC surfaces indicates that Ag and Au evaporated species in the Ar or N_2 plasma are transported unreacted to the oxidized surfaces. However, when Au and Ag are evaporated in the $(\text{Ar} + \text{O}_2)$ plasma, it has been reported^{13,14} that Au and Ag undergoes a reaction to form a metastable gold oxide Au_2O_3 or silver oxide Ag_2O on the surface. A reaction of the depositing Au or Ag species within the oxidizing plasma lead to a reactive deposition of a metastable-oxide compound which favorably affects the nucleation and interface formation. The CTE mismatch as shown in Table 1 between the noble metals and the carbides and nitrides should be reduced due to the reactively sputter-oxidized Si_3N_4 and SiC surfaces and the oxidized-metastable Ag and Au compound formation during deposition. And further, the activated-oxidized-metastable Ag and Au species may more effectively “wet” the oxidized surface due to an increased contact area.¹⁵

Friction Testing

The pin-on-disk friction results of oxygen assisted SCIP'd Ag and Au films run at a atmospheric conditions under 4.9 N load at 1 m/min (370 rpm) sliding velocity for 30 min at 25 °C are tabulated in Figs. 6(a) and (b). Unlubricated Si_3N_4 and SiC disks displayed high friction coefficient of 0.72 and 0.85 respectively. The Au SCIP'd films on Si_3N_4 and SiC surfaces displayed the lowest friction coefficients of 0.15 and 0.19 respectively. The Ag SCIP'd films on Si_3N_4 and SiC displayed a coefficient of friction of 0.30 and 0.38 respectively. Thus the Au films reduced the coefficient of friction by a factor of 4, whereas the Ag films reduced it by a factor of 2.5

Conclusions

1. The Screen Cage Ion Plating (SCIP) process enables one to plate metals on electrically nonconductive ceramics (Si_3N_4 , SiC) with a three-dimensional coverage.
2. A dramatic increase in Ag or Au film adherence to Si_3N_4 and SiC surfaces is achieved when SCIP is performed in a reactive 50% O_2 + 50%Ar glow discharge.
3. Oxidation transformed Si_3N_4 and SiC surfaces and the formation of activated-oxidized-metastable Ag and Au species in the glow discharge should cooperatively contribute to the excellent film adherence.
4. The thermal expansion mismatched between the film and substrate should be reduced at the interface.
5. The Ag and Au plated films on Si_3N_4 and SiC surfaces reduce the sliding friction coefficient by a factor of 2-1/2 to 4.

ACKNOWLEDGMENT

The author would like to thank Dr. D. Wheeler for the XPS analysis.

REFERENCES

1. C. DellaCorte, F.S. Honey, and S.V. Pepper, "Tribological Properties of Ag/Ti Films on Al_2O_3 Ceramic Substrates," NASA TM-103784, 1991.
2. P.A. Benoy and C. DellaCorte, "Tribological Characteristics of Sputtered Au/Cr Films on Alumina Substrates Elevated Temperatures," *Surface and Coating Technology*, 62 (1993), pp. 454-459.
3. Kwang Soo Yoo, "Adhesion, Surface Morphology, and Gas Sensing Characteristics of Thin-Gold-Film Chemical Sensor," *J. Vac. Sci/Technical A* 92(1), Jan/Feb. 1984, pp. 192-198.
4. K. Schneider and H.W. Grunling, "Mechanical Aspects of High Temperature Coatings," *Thin Solid Films*, 107, (1983), pp. 395-416.
5. L.C. Erickson, et al., "Tribological Characterization of Alumina and Silicon Carbide Under Lubricated Sliding," *Tribology International*, Vol. 26, No. 2, 1993, pp. 83-92.
6. F.P. Bowden and D. Tabor, "Friction and Lubrication of Solids," Part 1, Clarendon Press, Oxford (1950) III.
7. D.H. Buckley, "Surface Effects in Adhesion, Friction, Wear and Lubrication," Elsevier (1981), p. 324.
8. T. Spalvins and H.E. Sliney, "Screen Cage Ion Plating (SCIP) and Scratch Testing of Polycrystalline Aluminum Oxide," NASA TM-105404, 1992.
9. T. Spalvins and H.E. Sliney, "Frictional Behavior and Adhesion of Ag and Au Films Applied to Al_2O_3 , by Oxygen-Ion Assisted Screen Cage Ion Plating," *Surface and Coatings Technology*, 68/69 (1994) pp. 482-488.
10. H.E. Sliney and C. DellaCorte, "The New Test Machine for Measuring Friction and Wear in Controlled Atmosphere at 1200 °C," *Lubrication Engineering*, Vol. 47, No. 4, April 1991, pp. 314-319.
11. L.U.T. Ogbuji and D.T. Jayne, "Mechanisms of Incipient Oxidation of Bulk Chemical Vapor Deposited Si_3N_4 ," *J. Electrochem. Soc.*, Vol. 140, No. 3, March 1993, pp. 759-766.
12. L.C. Erickson et al., "Tribological Characterization of Alumina and Silicon Carbide Under Sliding Conditions," *Tribology International*, Vol. 26, No. 2, 1993, pp. 83-92.
13. L. Maya and M. Paranthaman, "Gold Oxide as Precursor to Gold/Silica Nanocomposites," *J. Vac. Sci. Technol*, B14 (1) Jan/Feb 1996, pp. 15-21.
14. L. Maya, "Gold Nanocomposites," *J. Vac. Soc. Technol B* 13(2), Mar/Apr 1995, pp. 362-365.
15. Takahashi and O. Kuboi, "Study on Contact Angles of Au, Ag, Cu, Sn, Al and Al alloys to SiC," *J. of Materials Science*, 34, (1996); pp. 1797-1802.

Table 1.—Linear Thermal Expansion Coefficients
 $(\alpha) \times 10^{-6} \text{ }^{\circ}\text{C}^{-1}$

Al_2O_3 —7.1
Si_3N_4 —3.1
SiC—4.5
Ag—19.6
Au—14.2

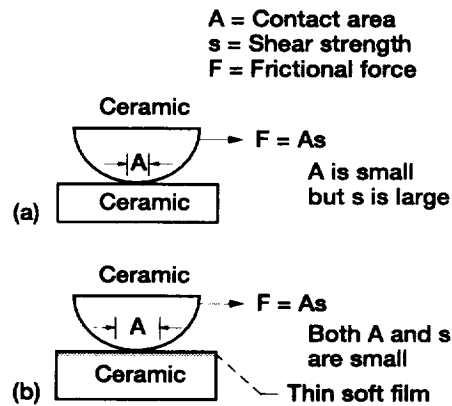


Fig. 1.—Principles of thin film lubrication.

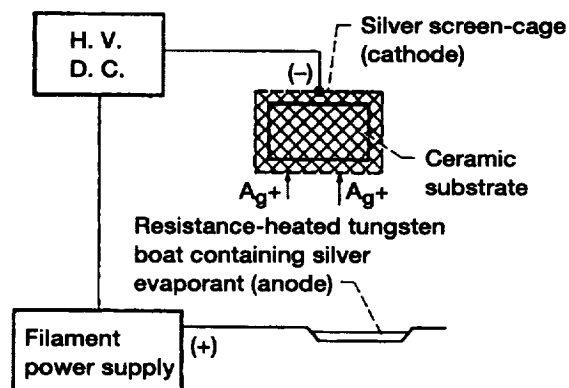


Fig. 2.—Schematic for screen cage ion plating of ceramics.

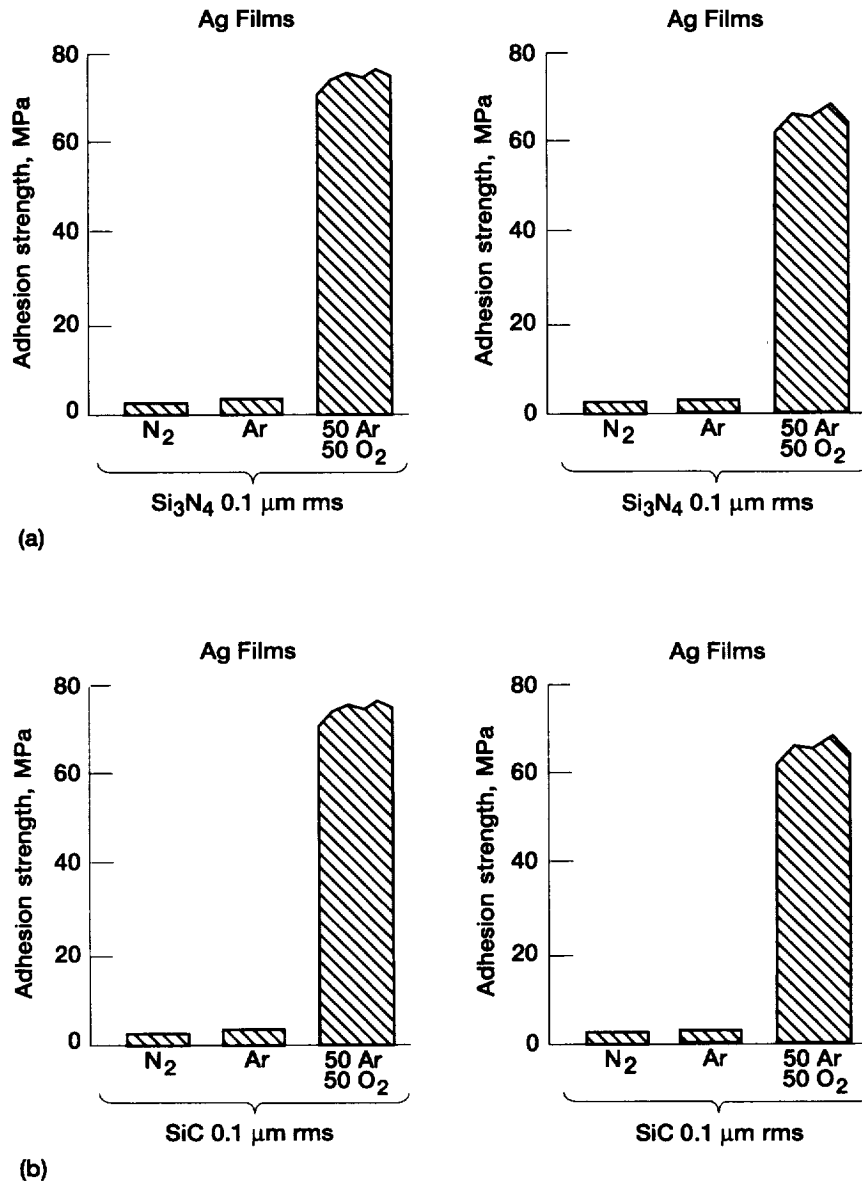


Fig. 3.—Adhesion strengths of SCIP'd Ag and Au films at different plasma compositions: (a) on Si₃N₄; (b) on SiC.

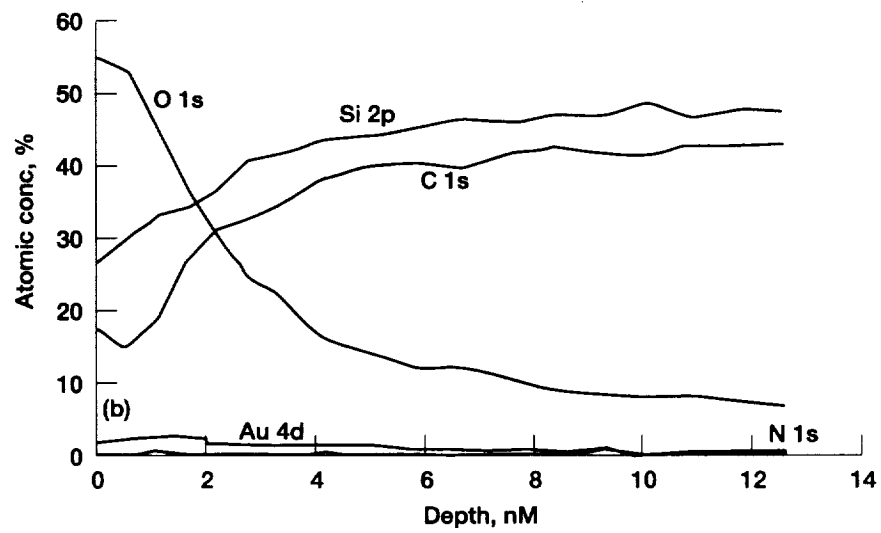
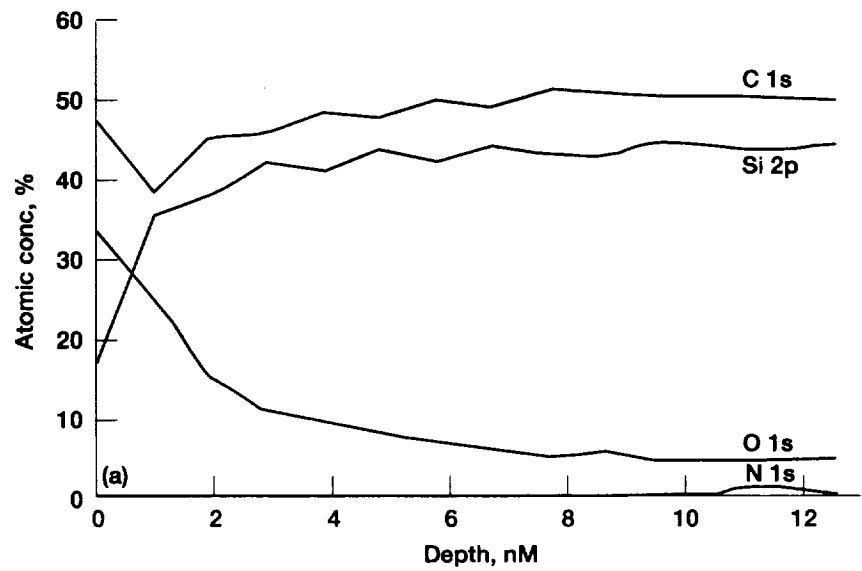


Fig. 4.—XPS depth profiles of SiC after argon and after an oxygen sputter etch in Screen cage: (a) Ar etch; (b) (Ar + O₂) etch.

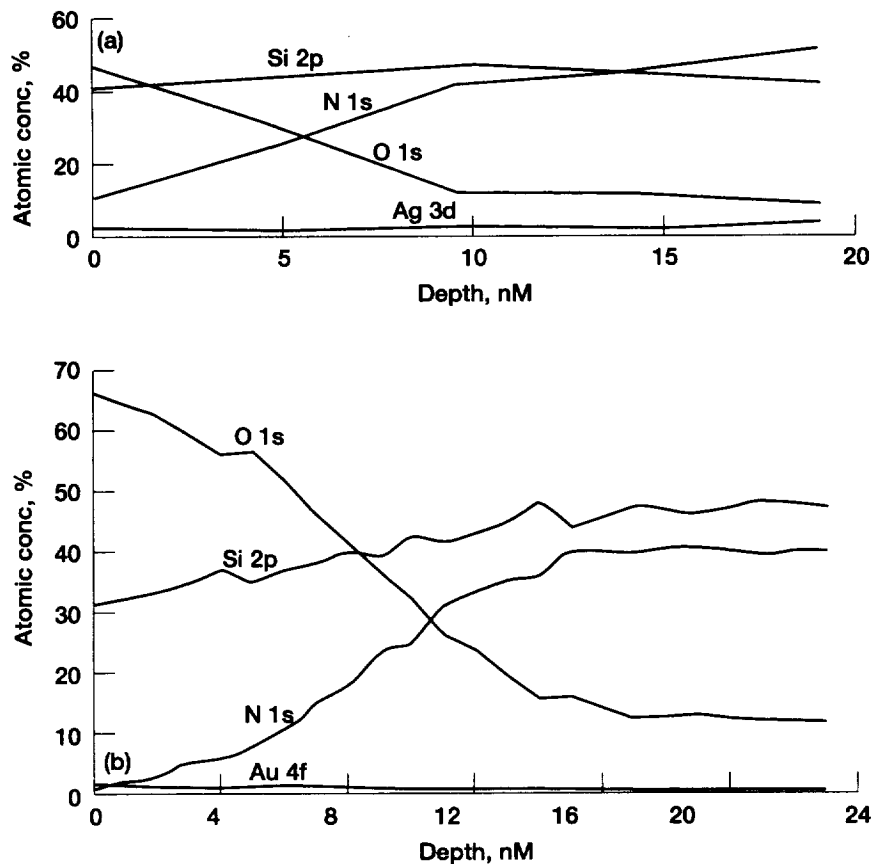


Fig. 5.—XPS depth profiles of Si_3N_4 after argon and after argon/oxygen sputter etching in screen cage: (a) Ar etch; (b) (Ar + O₂) etch.

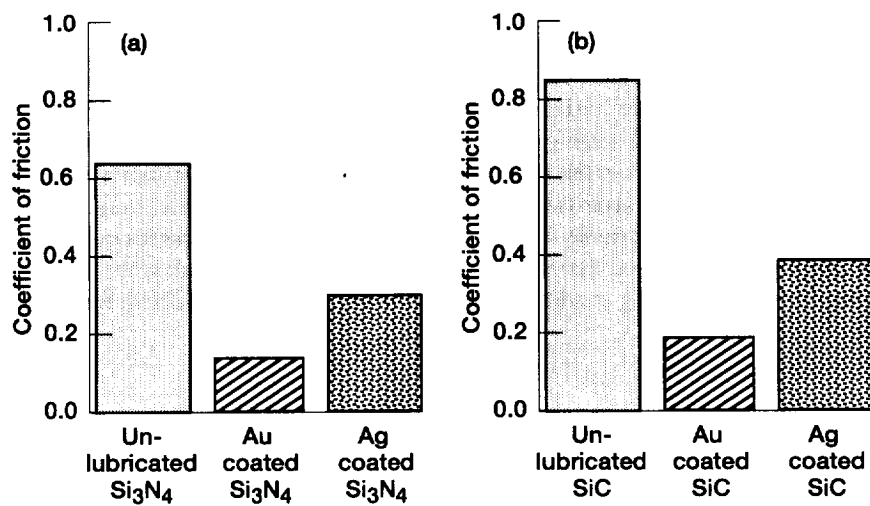


Fig. 6.—Friction coefficient of Ag and Au reactively SCIP'd films: (a) on Si_3N_4 substrates; (b) on SiC substrates.

REPORT DOCUMENTATION PAGE			Form Approved OMB No. 0704-0188	
Public reporting burden for this collection of information is estimated to average 1 hour per response, including the time for reviewing instructions, searching existing data sources, gathering and maintaining the data needed, and completing and reviewing the collection of information. Send comments regarding this burden estimate or any other aspect of this collection of information, including suggestions for reducing this burden, to Washington Headquarters Services, Directorate for Information Operations and Reports, 1215 Jefferson Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management and Budget, Paperwork Reduction Project (0704-0188), Washington, DC 20503.				
1. AGENCY USE ONLY (Leave blank)	2. REPORT DATE June 1998	3. REPORT TYPE AND DATES COVERED Technical Memorandum		
4. TITLE AND SUBTITLE The Improvement of Ion Plated Ag and Au Film Adherence to Si ₃ N ₄ and SiC Surfaces for Increased Tribological Performance		5. FUNDING NUMBERS WU-523-22-13-00		
6. AUTHOR(S) Talivaldis Spalvins				
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) National Aeronautics and Space Administration Lewis Research Center Cleveland, Ohio 44135-3191		8. PERFORMING ORGANIZATION REPORT NUMBER E-11169		
9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES) National Aeronautics and Space Administration Washington, DC 20546-0001		10. SPONSORING/MONITORING AGENCY REPORT NUMBER NASA TM-1998-207415		
11. SUPPLEMENTARY NOTES Responsible person, Talivaldis Spalvins, organization code 5140, (216) 433-6060.				
12a. DISTRIBUTION/AVAILABILITY STATEMENT Unclassified - Unlimited Subject Category: 27 This publication is available from the NASA Center for AeroSpace Information, (301) 621-0390.			12b. DISTRIBUTION CODE	
13. ABSTRACT (Maximum 200 words) A modified dc-diode plating system, utilizing a metallic screen cage as a cathode and referred as SCREEN CAGE ION PLATING (SCIP), is used to deposit Ag and Au lubricating films on Si ₃ N ₄ and SiC surfaces. When deposition is performed in Ar or N ₂ glow discharge, the surface displays poor adhesive strength (<5 MPa). A dramatic increase in adhesive strength (>80 MPa) is achieved when plating is performed in a reactive 50% O ₂ + 50% Ar glow discharge. The excited/ionized oxygen species (O ₂ ⁺ /O ⁺) in the glow discharge contribute to the oxidation of the Si ₃ N ₄ or SiC surfaces as determined by X-ray Photoelectron Spectroscopy (XPS) depth profiling. The reactively sputter-oxidized Si ₃ N ₄ or SiC surfaces and the activated-oxidized-metastable Ag or Au species formed in the plasma cooperatively contribute to the increased adherence. As a result, the linear thermal expansion coefficient mismatch at the interface is reduced. These lubricating Ag and Au films under sliding conditions reduce the friction coefficient by a factor of 2-1/2 to 4.				
14. SUBJECT TERMS Lubricants; Thin films; Ceramics			15. NUMBER OF PAGES 15	
			16. PRICE CODE A03	
17. SECURITY CLASSIFICATION OF REPORT Unclassified	18. SECURITY CLASSIFICATION OF THIS PAGE Unclassified	19. SECURITY CLASSIFICATION OF ABSTRACT Unclassified	20. LIMITATION OF ABSTRACT	